

# Carbon derived from Jute Sacks For Low-Cost and Efficient Counter Electrodes of Dye-Sensitized Solar Cells

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Non-activated and activated carbons derived from jute sacks were used as counter electrode (CE) materials of dye-sensitized solar cells (DSSCs). The non-activated carbon sample was synthesized by carbonization of jute fibers at 600°C in N<sub>2</sub>. The activated carbon samples were synthesized by carbonizing the ZnCl<sub>2</sub> activated jute at 600 and 800°C in N<sub>2</sub>. The x-ray diffraction revealed that the jute-based carbon samples comprised of graphitic and amorphous carbons. Raman spectroscopy disclosed that the disordered carbon was dominant in all samples. Based on the energy dispersive x-rays spectroscopy measurements, activated carbon prepared at 800 °C had the highest amount of carbon content (~94.3%). The surface textural properties of the samples were investigated with N<sub>2</sub> adsorption-desorption experiment. Brunauer-Emmett-Teller surface area of non-activated (NACJ6), activated at 600 °C (ACJ6) and activated at 800 °C (ACJ8) carbon samples were 23.73, 1912, and 1613 m<sup>2</sup>g<sup>-1</sup>, respectively, and their corresponding pore volumes were 0.02448, 1.114 and 0.9625 cm<sup>3</sup>g<sup>-1</sup>, respectively. ACJ8 with the highest carbon content, moderately high surface area, and micropores and mesopores/macropores demonstrated superior catalytic ability among the samples. The efficiency of the light to electric power conversion of the DSSCs using ACJ8, ACJ6, and NACJ6 as CEs were found to be about 3.35, 2.75, and 0.88%, respectively, compared to 3.39% that of the DSSC using platinum (Pt). Obtained efficiency comparable to Pt based DSSC using the ACJ8 sample implies that the jute-based activated carbon can be an inexpensive and efficient CE material alternative to Pt for commercialization.

**Keywords:** Activated carbon; Counter electrode; Catalyst; DSSCs; Iodide/tri-iodide ions; Jute; Platinum

## 1 Introduction

Dye-sensitized solar cells (DSSCs) are considered as cost-effective alternative devices to conventional solar cells<sup>1</sup>. A DSSC is an assembly of photoelectrode and counter electrode (CE) with an electrolyte encapsulated between the electrodes. The photoelectrode is the front electrode, typically consists of a fluorine doped tin oxide (FTO)-glass substrate coated with a film of high band gap mesoporous nanoparticles such as titanium dioxide (TiO<sub>2</sub>). A monolayer of a dye molecules is adsorbed on the TiO<sub>2</sub>. The dye molecules inject their electrons into TiO<sub>2</sub> film as they receive sunlight. The photoelectrons diffuse in the TiO<sub>2</sub> film and move to the CE via an external circuit. Generally, a CE is an FTO-glass substrate coated with a catalyst like

platinum (Pt). The dye molecules are regenerated by receiving electrons from iodide ions in the electrolyte. After donating electrons, iodide ions become tri-iodide ions which migrate toward the CE. The tri-iodide ions receive electrons at the CE and turns to iodide ions. The Pt coated on the CE enhances the conversion of tri-iodide into iodide ions<sup>1-11</sup>. However, high price of Pt motivated researchers to develop low-cost alternate materials for CEs. Carbonaceous materials like activated carbon<sup>8,12-18</sup>, carbon black<sup>1,19,20</sup>, carbon nanotubes<sup>21-23</sup>, carbon nanofibers<sup>5,7,24</sup>, graphene<sup>25,26</sup>, and lampblack<sup>9-11</sup> have been employed as CE materials alternative to Pt.

Among the carbonaceous materials, the activated carbon is promising because of their high surface area and porosity<sup>14</sup>. Additionally, activated carbons can be synthesized from readily available various types of biomasses using simple method<sup>27-31</sup>. Thus, biomass-

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based activated carbon can be a low-cost CE material for DSSCs. Catalytic ability of a carbonaceous material is affected by its surface textural properties like surface area<sup>14</sup>, porosity and pore size distribution<sup>32</sup>. Also, the purity of CE material may be another influential factor. Surface textural properties and structural composition of the biomass-based carbon are greatly influenced by activation process and carbonization temperature<sup>27–29,31</sup>. In this report, we prepared activated carbons from jute sacks, and employed them as CE material for DSSCs. Generally, jute sacks are used for packing commodities and used jute sacks are disposed as a waste. We have utilized waste sacks as it is readily available and hence a low-cost precursor for the preparation of the carbonaceous catalyst. Three different types of carbon samples: non-activated, activated at 600 °C and activated at 800 °C were derived from the jute sacks, and characterized for their surface texture, porosity, structural and chemical composition properties to study the influence of the activation conditions. The carbon samples were used as counter electrode (CE) materials of dye-sensitized solar cells (DSSCs) to study their catalytic ability and found that the solar cell with the activated carbon carbonized at 800 °C exhibited superior photovoltaic performance with the efficiency comparable to the efficiency of the reference solar cell using Pt.

## 2 Experimental Method

### 2.1 Preparation of Non-Activated and Activated Carbons

Slices of used jute sacks were boiled in distilled water with detergent and jute fibers were extracted from them. The jute fibers were used as precursors for the preparation of carbons. The method of preparation of the carbons is similar to that described elsewhere<sup>27,28,31</sup>. The non-activated carbon was prepared by heating the precursor in a quartz tube furnace (Zhengzhou Protech Technology Co., LTD,

China) at 600 °C in N<sub>2</sub> for 4 hours. To prepare the activated carbon samples, the jute fibers were soaked in aqueous solution of ZnCl<sub>2</sub> for 24 hours. Equal amounts of jute fibers and ZnCl<sub>2</sub> by weight were taken for the activation process. The activated jute fibers were dried and carbonized at two different temperatures at 600 and 800 °C in N<sub>2</sub>. The carbon samples were washed with dilute HCl and distilled water successively. Then the samples were dried, ground, and filtered (through 120-micron sieve). The non-activated carbon carbonized at 600 °C, the activated carbon carbonized at 600 °C, and the activated carbon carbonized at 800 °C, were labelled as NACJ6, ACJ6, and ACJ8, respectively.

### 2.2 Characterization of Carbon Samples

The X-ray diffraction (XRD) measurements of the samples were conducted for structural characterization using a Bruker D2 PHASER diffractometer ( $\lambda = 1.54 \text{ \AA}$ ). Raman spectroscopy measurements were performed by using Laser of 532 nm wavelength. Scanning electron microscopy (SEM) and energy dispersive X-ray spectroscopy (EDS) were performed using FEI Helios Nanolab 400 Scanning Electron Microscope.

Surface textural parameters like pore volume and pore size distribution (PSD) of activated carbon (catalyst) influence the transport mechanism of electrolyte<sup>33</sup> and hence its catalytic ability. Surface textural properties of the non-activated and activated carbons were investigated by N<sub>2</sub> adsorption and desorption experiments<sup>34,35</sup> using Autosorb-1C, Quantachrome.

### 2.3 Fabrication of Dye-Sensitized Solar Cells

DSSCs with CEs coated with NACJ6, ACJ6, and ACJ8 samples were prepared. A DSSC with Pt CE was also prepared as a reference solar cell. The structure of a DSSC with carbon-based CE is shown in Figure 1. The procedure of the preparation of the

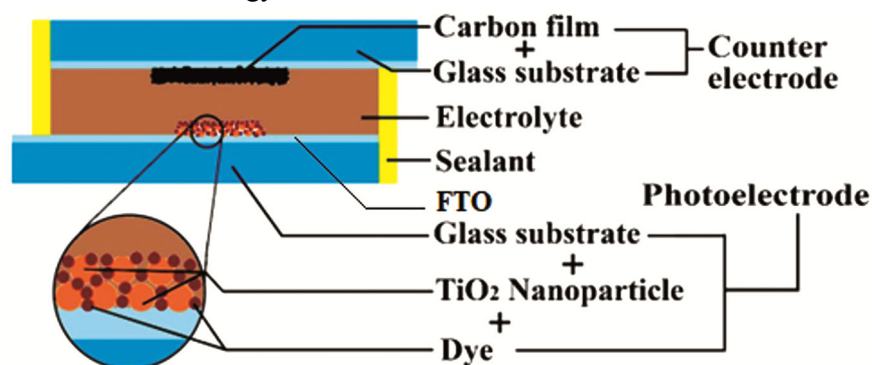


Fig. 1 — Schematic diagram of a cross-section of a DSSC with a carbon-based

carbon-based CEs was similar to the previous reports<sup>5,6</sup>. Carbon paste was prepared by mixing the carbon sample with an aqueous solution of carboxy methylcellulose sodium salt (CMC) of 2% concentration. The CMC was added as a binder. The carbon paste was doctor bladed onto FTO-glass substrates and dried overnight in an oven at 70 °C. Pt based CEs were prepared by coating its precursor (Plati Sol, Solaronix) onto an FTO-glass substrates and then heating the precursor at 450 °C for 45 minutes.

The photoelectrodes of the DSSCs were prepared as follows<sup>8,9</sup>. A thin layer of nano-crystalline TiO<sub>2</sub> paste (TiNanoxide T/SP, Solaronix) was doctor bladed onto FTO-glass substrates. The TiO<sub>2</sub> paste was sintered at 450 °C for 45 minutes and then cooled down to ~80 °C. The sintered TiO<sub>2</sub> film was soaked in 0.5 mM N-719 dye (Ruthenizer 535-bisTBA, Solaronix) solution for 24 hours. The photoelectrode and counter electrode were assembled with a paraffin film which acts as a sealant and spacer. Electrolyte (Idolyte AN-50, Solaronix) containing iodide and triiodide ions was injected into the gap between the electrodes. A DSSC with Pt coated CE was also fabricated as a reference solar cell<sup>9</sup>.

#### 2.4 Photovoltaic Characterization of DSSCs

The photovoltaic performance of the DSSCs was characterized with the current density-voltage (J-V) characteristics<sup>6,36</sup> of the solar cells under the illumination of ~100 mW/cm<sup>2</sup> from a solar simulator (Abet technologies, model 11002). Source meter (Ossilla, UK) was employed to obtain the J-V curves. Photovoltaic parameters such as light- to-electricity conversion efficiency ( $\eta$ ), short-circuit current density ( $J_{sc}$ ), open circuit voltage ( $V_{oc}$ ), fill factor (FF), and series resistance ( $R_s$ ) of the solar cells were calculated from the J-V curves<sup>6,36</sup>. An application software provided with the source meter (Ossilla) was used to determine the values of the photovoltaic parameters.

### 3 Results and discussion

#### 3.1 XRD and Raman spectroscopy

Figure 2 shows the XRD patterns of the carbon samples with two broad and characteristic peaks due to reflections from (002) and (100) plans centered around  $2\theta = 24.96^\circ$  and  $43.64^\circ$  from NACJ6 (Fig. 2(a)),  $2\theta = 25.05^\circ$  and  $43.10^\circ$  from ACJ6 (Fig. 2(b)), and  $2\theta = 26.57^\circ$  and  $43.69^\circ$  from ACJ8 (Fig. 2(c)). These broad characteristic peaks indicate the carbon samples contain graphitic crystallites<sup>29,37,38</sup>. Moreover, the XRD patterns of the carbons show large background

in the region of (002) peaks<sup>39</sup>. According to previous reports<sup>39,40</sup>, such background signal is attributed to the reflection from amorphous carbon. Lu *et al.*<sup>41</sup> attributed that the intensity of (002) peak is the sum of the intensity scattered from crystalline carbon and amorphous carbon<sup>39,41</sup>. Based on the XRD patterns of the activated carbons, the intensity contributed from the amorphous carbon seems to be much higher than that from the crystalline carbon indicating significantly high fraction of amorphous carbon present compared to the crystalline carbon in these samples. Also, the XRD-pattern of ACJ6 (Fig. 2(b)) shows sharp peaks at about  $2\theta = 31.63, 36.44, 47.66, 56.47, 62.68$  and  $68.09^\circ$ . The peaks correspond to the reflection from (100), (101), (102), (110), (103), and (112) planes of ZnO<sup>42,43</sup>. Gnawali *et al.*<sup>31</sup>, while reporting the preparation of carbon from ZnCl<sub>2</sub> activated *Terminalia chebula* seed for super capacitor application, mentioned that the ZnCl<sub>2</sub> is partially converted into solid ZnO at higher pyrolysis temperature above 290 °C. The low intensity ZnO peaks seen in the XRD pattern of ACJ6, indicate that ZnO still remained as impurity even after washing the sample with HCl.

Figure 3 shows the Raman spectra of NACJ6, ACJ6 and ACJ8 samples. Raman spectrum of all samples showed two distinct peaks centered at 1347.09 and 1588.60 cm<sup>-1</sup> for NACJ6, at 1346.38 and 1589.99 cm<sup>-1</sup> for ACJ6, and at 1345.92 and 1589.49 cm<sup>-1</sup> for ACJ8 samples. The peak 2 in the samples at 1588.60, 1589.99 and 1589.49 cm<sup>-1</sup> correspond to the G-band which is omnipresent in all the carbonized samples<sup>28,44</sup>. This indicates the presence of ordered graphitic structure which is attributed to E<sub>2g</sub> mode vibrations of graphitic planes. Additionally, the peak 1 in the samples at 1347.09,

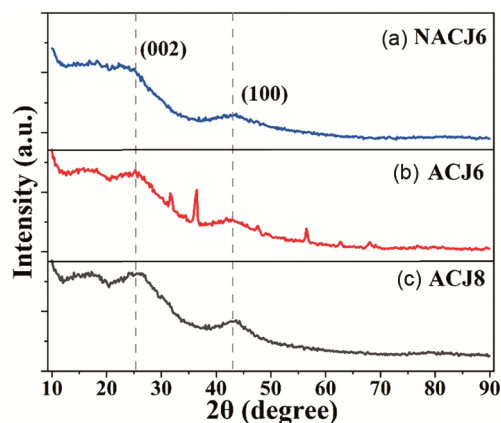


Fig. 2 — XRD pattern of (a) NACJ6, (b) ACJ6, and (c) ACJ8

1346.38, and 1345.92  $\text{cm}^{-1}$  correspond to the D-band which indicates the presence of disordered carbon structure<sup>5,28,29,44</sup>. The Raman spectra were fitted with Lorentzian line shapes to calculate the portions of the ordered and disordered carbon phases<sup>45</sup>. The NACJ6 sample contains about 76.94% disordered and 23.05% ordered carbon. Similarly, the ACJ6 sample contains about 75.00% disordered and 24.99% ordered carbon, and the ACJ8 sample contains about 74.36% disordered and 25.63% ordered carbon. This shows that amorphous carbon dominates the crystalline form of the carbon in all samples, however, degree of crystallinity increased slightly as carbonization temperature is increased from 600 to 800 °C. These results of Raman spectroscopy of the carbon samples correlate with their XRD patterns.

### 3.2 SEM images and EDS

Figure 4 shows the SEM images of the carbon samples. Fig. 4(a-c) are the SEM images of NACJ6,

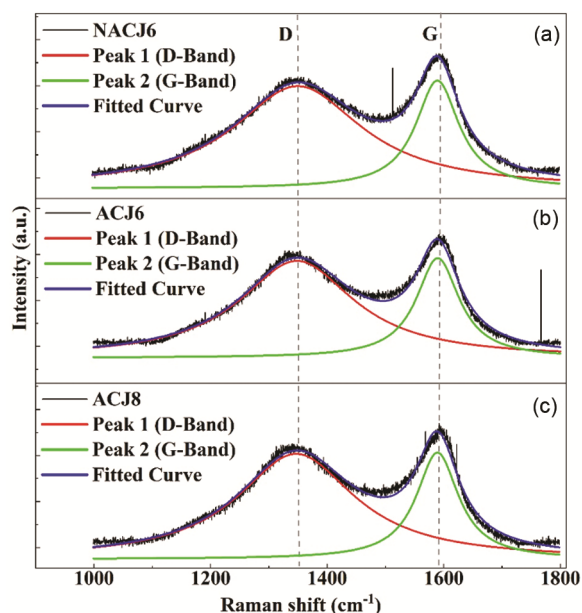


Fig. 3 — Raman spectrum of (a) NACJ6, (b) ACJ6, and (c) ACJ8

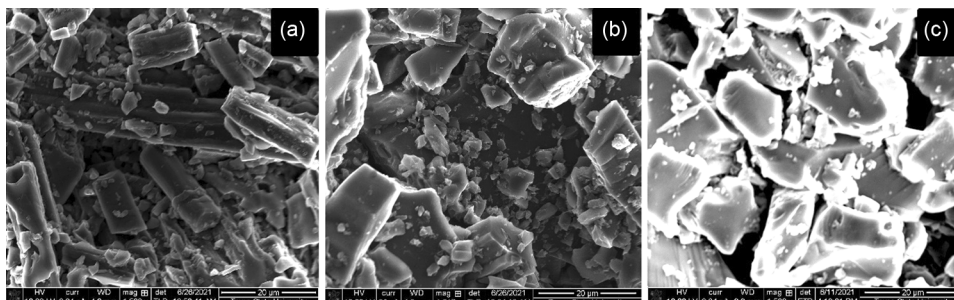


Fig. 4 — SEM image of (a) NACJ6, (b) ACJ6, and (c) ACJ8

ACJ6, and ACJ8 samples, respectively. The size of the scale bar in each SEM image is 20  $\mu\text{m}$ . The SEM image of NACJ6 shows that the carbon sample consists of irregular shaped particles. Solid cylindrical shaped carbon particles are dominant ones and the size of the particles ranges from few microns to 60  $\mu\text{m}$ . The SEM images of ACJ6 and ACJ8 show that the particles are mostly granular in shape and their sizes are less than 20  $\mu\text{m}$ . It seems that the carbonization of the activated precursors yielded smaller particles compared to the carbonization of the precursor without activation. Generally, smaller particles yield larger surface area.

Figure 5 shows the SEM-EDS spectra of NACJ6, ACJ6, and ACJ8. The scanned area of EDS of each

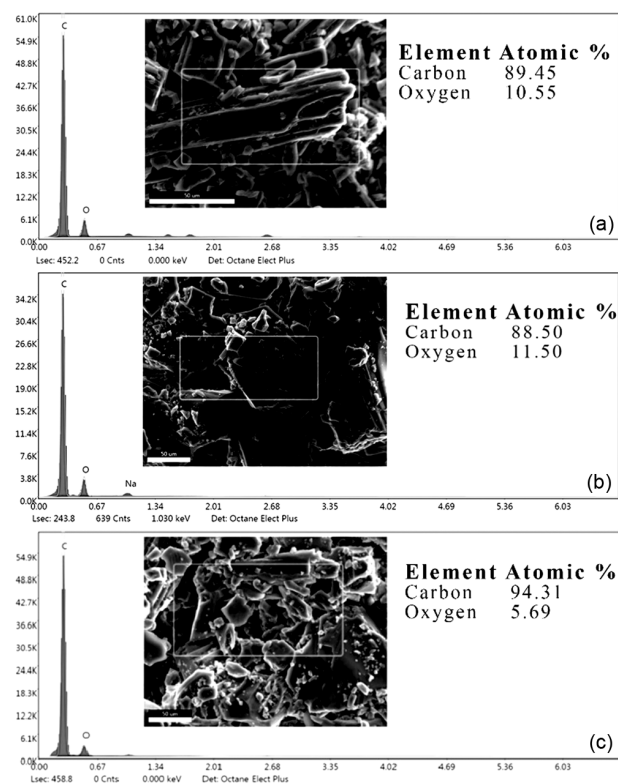


Fig. 5 — EDS spectrum of (a) NACJ6, (b) ACJ6, and (c) ACJ8

sample is enclosed within a square in its SEM image shown in inset. Based on the EDS results, the carbon content in NACJ6, ACJ6, and ACJ8 are about 89.5, 88.5, and 94.3%, (atomic %), respectively. The EDS revealed that carbon content in NACJ6 and ACJ6 are almost equal, but significantly high in ACJ8 which is also consistent with higher integrated intensity of the peaks seen in the XRD pattern of ACJ8 (compared to those of NACJ6 and ACJ6). This indicates that higher carbonization temperature is more favorable for reducing impurities in the biomass derived carbon samples.

### 3.3 Surface Textural Properties

The  $N_2$  adsorption-desorption isotherms of NACJ6, ACJ6, and ACJ8 are shown in Fig. 6. The  $N_2$  adsorption-desorption isotherms were recorded by varying relative pressure ( $P/P_0$ ) from  $\sim 0.05$  to 0.99 at

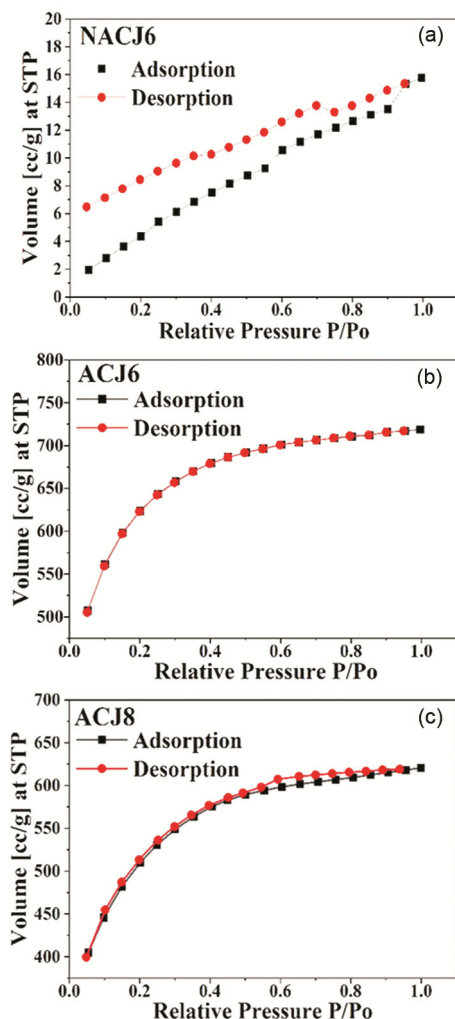


Fig. 6 —  $N_2$  adsorption-desorption isotherm plots for (a) NACJ6, (b) ACJ6, and (c) ACJ8

bath temperature of 77.3 K, where  $P$  is pressure of  $N_2$  (adsorptive) and  $P_0$  is saturation pressure. The isotherm of NACJ6 shows the adsorption of  $N_2$  increases linearly with increasing  $P/P_0$ . The linear section of the isotherm indicates formation of multilayer of  $N_2$  on the surface of the carbon. This isotherm belongs to Type II and it is a characteristic of a non-porous or macroporous solid<sup>34</sup>. Additionally, the isotherm shows that the volume of  $N_2$  adsorbed by this carbon at the entire range of the gas pressure is extremely small implying its low surface area. In contrast, the isotherms of ACJ6 and ACJ8 are nonlinear. The  $N_2$  adsorption increases steadily above  $\sim 0.5 P/P_0$  and approaches a limiting value at  $\sim 0.99 P/P_0$ . The volume of  $N_2$  adsorbed on carbon surfaces are notably higher at all pressure indicating its surface area is significantly high compared to that of NACJ6. For ACJ6, the isotherm is of Type-I<sup>34,35</sup> and it indicates that ACJ6 is a mostly microporous carbon<sup>34,37</sup>. Pore sizes in materials are categorized into micropores ( $< 2$  nm), mesopores ( $2 \text{ nm} < 50 \text{ nm}$ ), and macropores ( $> 50 \text{ nm}$ )<sup>34</sup>. In case of ACJ8, its isotherm shows a weak hysteresis loop at  $\sim 0.5$  to  $0.8 P/P_0$  due to capillary condensation of the adsorbate in mesopores. This isotherm belongs to Type IV and it indicates the formation of significant number of mesopores along with dominant micropores in this carbon<sup>29,34,37</sup>. The isotherm also depicts that adsorption of less amount of  $N_2$  on ACJ8 compared to ACJ6 indicating its smaller specific surface area compared to that of later one.

Figure 7 shows the PSD profiles of NACJ6, ACJ6, and ACJ8 obtained by Density Function Theory (DFT) method<sup>29,33,46</sup>. The PSD profiles reveal that NACJ6 is the least porous whereas ACJ6 and ACJ8 are highly porous. Also, the profiles depict the presence of higher portion of micropores in ACJ6 and mesopore in ACJ8. As the size of iodide/tri-iodide ions in the electrolyte used in DSSCs are of a few angstroms in dimensions<sup>12</sup>, the pore sizes in the activated carbons are sufficiently larger for easy movement of the ions through the pores.

The surface textural parameters--BET specific surface area ( $S_{BET}$ ), total pore volume ( $V_{tot}$ ), and micropore area ( $S_{micro}$ )-- obtained from the  $N_2$  adsorption-desorption experiments are as follow. The  $S_{BET}$ ,  $V_{tot}$ , and  $S_{micro}$  of NACJ6 are  $23.73 \text{ m}^2\text{g}^{-1}$ ,  $0.02448 \text{ cm}^3\text{g}^{-1}$ , and  $11.54 \text{ m}^2\text{g}^{-1}$ , respectively. These parameters of ACJ6 are  $1912 \text{ m}^2\text{g}^{-1}$ ,  $1.114 \text{ cm}^3\text{g}^{-1}$ , and  $1755 \text{ m}^2\text{g}^{-1}$ , respectively. Similarly, the corresponding values of the parameters of ACJ8 are  $1613 \text{ m}^2\text{g}^{-1}$ ,  $0.9625 \text{ cm}^3\text{g}^{-1}$ , and  $1443 \text{ m}^2\text{g}^{-1}$ , respectively.

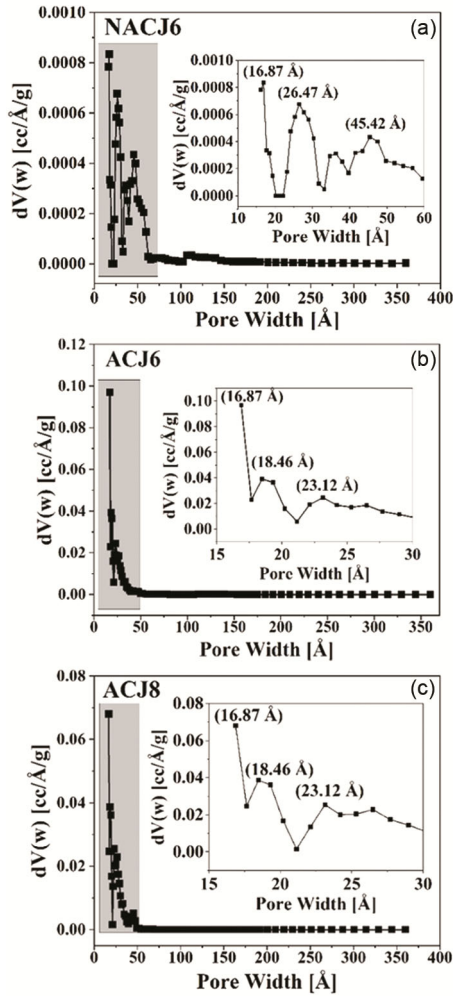


Fig. 7 — Pore size distributions (PSD) obtained from DFT methods for (a) NACJ6, (b) ACJ6, and (c) ACJ8

The specific surface area of the activated carbon samples significantly increased (at least by ~68 times) compared to that of the non-activated carbon because of increase in their porosity. However, the specific surface area of the activated carbon reduced as carbonization temperature was elevated from 600 to 800 °C. This could be due to merge of some of the micropores forming more mesopores during the carbonization at the elevated temperature<sup>29</sup>. The micropore surface area was calculated by t-method (de Boer method)<sup>47</sup>. The cumulative value of mesopore and macropore area (named  $S_{\text{meso/macro}}$ ) was calculated by subtracting the micropore area from the total specific surface area ( $S_{\text{BET}}$ ). The  $S_{\text{meso/macro}}$  of NACJ6, ACJ6, and ACJ8 are 12.19, 157.0, and 170.0  $\text{m}^2\text{g}^{-1}$ , respectively. The ratio of mesopore/macropore to micropore surface area of ACJ8 is 0.12 and that of ACJ6 is 0.09. Thus, the

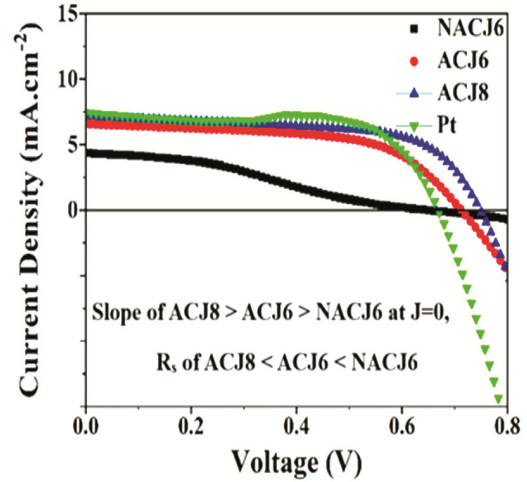


Fig. 8 — J-V Curves of DSSCs with CEs based on NACJ6, ACJ6, ACJ8, and Pt

elevated value of the ratio can be evidence of coalescence of micropores and formation of more mesopores/macropores. These results correlate with the appearance of hysteresis loop in the isotherm and the elevated volume of mesopores seen in the PSD profile of ACJ8.

### 3.4 Current density-Voltage Characteristics of Dye-Sensitized Solar Cells

Figure 8 shows the J-V curves of the DSSCs with CEs prepared using NACJ6, ACJ6, ACJ8, and Pt. The J-V curves show that the photocurrent from the DSSC with NACJ6 is the lowest and it is notably high from the DSSC with ACJ6. And the solar cell with ACJ8 produced the highest photocurrent which matched to that from the Pt based DSSC. Also,  $V_{\text{oc}}$  is the largest from the DSSC with ACJ8 while that for the devices with NACJ6 and Pt are nearly equal. Regarding the shape of the J-V curves, the J-V curve of NACJ6 shows least squareness and it improves for ACJ6 and ACJ8, successively. The shapes of J-V curves indicate low FF of the DSSC with NACJ6 and FF improved by notably for the solar cells with ACJ6 and ACJ8. Furthermore, the successive increment of the slope of the J-V curves around  $J=0$  for ACJ6 and ACJ8-based solar cells compared with that of NACJ6 indicates the decreasing trend of  $R_s$  of the activated carbon-based solar cells compared with the non-activated carbon-based solar cell.  $R_s$  of a solar cell can be estimated from the reciprocal of the slope of the linear part of the J-V curve at  $J=0$ <sup>6,48</sup>. The steepness of the J-V curves around  $J=0$  of ACJ6 is significantly large compared with that of NACJ6-based solar cell. And the slope of the J-V curve increases more for ACJ8-

based solar cell and almost matches to that of Pt-based solar cell. This implies that  $R_s$  of ACJ8 is the lowest among the carbon-based solar cells and nearly equal to that of the Pt-based solar cell.

The photovoltaic parameters  $\eta$ ,  $J_{sc}$ ,  $V_{oc}$ , FF, and  $R_s$  of the DSSC with NACJ6 are about 0.88 %, 4.37  $\text{mAcm}^{-2}$ , 0.64 V, 31.33 %, and 263.46  $\Omega\text{cm}^2$ , respectively. Those parameters for the solar cells with ACJ6 are about 2.75 %, 6.59  $\text{mAcm}^{-2}$ , 0.71 V, 58.70 %, and 21.02  $\Omega\text{cm}^2$ , respectively. Similarly, the values of the parameters for the DSSC with ACJ8 are about 3.35 %, 7.21  $\text{mAcm}^{-2}$ , 0.74 V, 62.31 %, and 12.77  $\Omega\text{cm}^2$ , respectively. And the corresponding values of the parameters for the solar cell with Pt are about 3.39 %, 7.45  $\text{mAcm}^{-2}$ , 0.66 V, 68.15 %, and 10.84  $\Omega\text{cm}^2$ , respectively. Table 1 lists the photovoltaic parameters of the DSSCs.

The  $\eta$  of the DSSC with ACJ6 is much larger than that of the solar cell with NACJ6. The  $R_s$  of NACJ6-based DSSCs is much larger compared to that of ACJ6 solar cell.  $R_s$  is a parasitic resistance in a solar cell and it dissipates electric power generated in the solar cell. Thus, higher value of  $R_s$  decreased FF<sup>36</sup> and  $J_{sc}$ <sup>12</sup> which caused low efficiency of NACJ6 solar cell. This indicates that catalytic ability of the non-activated carbon is significantly poor compared to the activated carbon. The elemental compositions of both of the carbons were similar but the surface area of the activated carbon was significantly larger than that of the non-activated carbon. The higher surface area of the catalyst provides larger catalyst-electrolyte interface for the reduction of tri-iodide ions in the electrolyte. This can be a major cause of superior catalytic ability of the activated carbon to the non-activated carbon. In previous study, Yoon *et al.*<sup>14</sup> adopted activated carbon derived from coconut shells, pine trees, and coals with BET areas of 1,111.32, 963.03, and 754.12  $\text{m}^2 \text{g}^{-1}$ , respectively, as CEs of the DSSCs and reported the efficiencies of 5.90, 6.21, 4.17 % from their respective solar cells compared to 7.20% efficiency from a reference cell with Pt. Their findings indicate carbonaceous CEs with high surface area exhibit better photovoltaic performance but the relation between increase in efficiency of the DSSCs and the surface area of the CE material is not linear one<sup>14</sup>.

Similarly, the  $\eta$  of the DSSC with ACJ8 was higher than that of the solar cell with ACJ6. Along with the efficiency, other photovoltaic parameters of ACJ8 were better than those of ACJ6. The EDS and PSD results of ACJ8 and ACJ6 showed that the variation of carbonization temperature results different carbon content and pore size profiles. ACJ8 with high purity provides more catalyst sites at the CE-electrolyte interface which is beneficial for expedite conversion of tri-iodide-ions into iodide ions compared to ACJ6. Regarding the PSDs, micropores were dominant pores in both types of the activated carbons, but the portion of mesopores/macropores was higher in the carbon activated at high temperature. Existence of abundant mesopores/macropores facilitates mass transport of electrolyte through the micropores<sup>28</sup>. This factor could have also enhanced the rate of the reduction of tri-iodide ions in the DSSC with ACJ8. Thus, among all the three types of the carbons based DSSCs, the solar cell with ACJ8 yielded the highest efficiency.

Moreover, the photovoltaic performance of the DSSCs with ACJ8 and Pt were almost same. The  $J_{sc}$  and FF of the ACJ8 based DSSC are slightly smaller than that of Pt based because of larger  $R_s$  (internal resistance) of the activated carbon-based solar cell compared to that of Pt based solar cell<sup>12</sup>. Smaller values of  $J_{sc}$  and FF of the ACJ8 based DSSC could reduce its  $\eta$  significantly compared with that of Pt-based device, however, elevated  $V_{oc}$  by about 80 mV of the former device helped to compensate the probable loss of its efficiency and enabled it to yield the efficiency which is comparable to that of Pt based solar cell. Imoto *et al.*<sup>12</sup> also reported the increase in  $V_{oc}$  by 60 mV of activated carbon (Bellfine AP-20, commercial activated carbon) based DSSC compared with sputtered Pt based DSSC. The authors attributed such increment of  $V_{oc}$  to the positive shift of the formal potential for iodide/tri-iodide ions in the electrolyte. The photovoltaic performance of the DSSCs with ACJ8 and Pt were almost same. The  $\eta$  of DSSC with ACJ8 is very close to that of the Pt-based DSSC. This shows that ACJ8 can be used for commercialization of DSSC with low-cost CE material alternative to Pt.

Table 1 — Photovoltaic parameters of DSSCs with CE based on NACJ6, ACJ6, ACJ8, and Pt

CE	$\eta$ (%)	$J_{sc}$ ( $\text{mAcm}^{-2}$ )	$V_{oc}$ (V)	FF (%)	$R_s$ ( $\Omega\text{cm}^2$ )
NACJ6	0.88	4.37	0.64	31.33	263.46
ACJ6	2.75	6.59	0.71	58.70	21.02
ACJ8	3.35	7.21	0.74	62.31	12.77
Pt	3.39	7.45	0.66	68.15	10.84

#### 4 Conclusion

Activated and non-activated carbon samples were synthesized from jute sacks aiming to employ them as Pt-free low-cost CEs in DSSCs. The carbon samples were characterized by XRD, Raman spectroscopy, SEM, EDS, and N<sub>2</sub> adsorption-desorption experiment for the exploration of their structural properties, chemical composition, and surface textural characteristics. We found the efficiency of the DSSC prepared with the activated carbon has significantly higher efficiency compared to non-activated carbon-based DSSC. Carbon samples activated at 800 °C possessed superior properties with the highest carbon content, moderately high surface area, and micropores and mesopores/macropores and demonstrated superior catalytic ability. The efficiency of the DSSC using the carbon sample activated at 800 °C as CE was found to be close to that obtained from the Pt-based reference DSSC. Hence, the activated carbon can be a novel, low-cost, and efficient CE material alternate to expensive Pt used in dye-sensitized solar cells.

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