

# Morphological studies of single phase alumina fabricated by wet milling method

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A wet milling process has been used as a novel method to fabricate single phase alumina. To fabricate the most thermodynamically stable form of alumina i.e.,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>, slurry method has been used here.  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> has the widest band gap  $\sim$ 8.8 eV which makes it more valuable in electronic devices for use as a perfect insulator. Further, to confirm the surface morphology of the prepared  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> sample, scanning electronic microscopy has been done. The prepared  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> sample has been characterized using X-ray diffraction pattern, Fourier transform infra-red spectroscopy and photoluminescence spectroscopy. Raman spectroscopy of prepared  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> has also been performed to understand the behaviour at microscopic level and also to confirm the purity of the single phase  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> sample.

**Keywords:** Single phase alumina,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>, Slurry, Wet milling

## Introduction

Alumina (Al<sub>2</sub>O<sub>3</sub>) is one of the highly used refractory oxides<sup>1-3</sup> and is utilized in a wide range of applications, including high-strength materials, in electronics<sup>4</sup>, ceramics, dielectric, optics<sup>5,6</sup>, mechanical engineering, and catalysts<sup>7-10</sup>. It is considered as a perfect insulator for various applications like gate dielectric in metal-oxide-semiconductor field effect transistor<sup>11-13</sup> to trapping the insulator in charge blocking non-volatile memory cells, due to its wide bandgap, low leakage of current and high value of dielectric under ambient conditions. Al<sub>2</sub>O<sub>3</sub> is known to exist in five metastable crystalline polymorphs<sup>14-16</sup> (transition aluminas), namely  $\gamma$ -Al<sub>2</sub>O<sub>3</sub><sup>17</sup>,  $\eta$ -Al<sub>2</sub>O<sub>3</sub>,  $\delta$ -Al<sub>2</sub>O<sub>3</sub>,  $\theta$ -Al<sub>2</sub>O<sub>3</sub>, and  $\chi$ -Al<sub>2</sub>O<sub>3</sub>. In addition to these,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>, also known as corundum or sapphire<sup>18,19</sup>, is considered as one of the thermodynamically stable phases<sup>20</sup>. Among the various forms, two phases such as,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub><sup>21</sup> are of great interest due to their technological applications. It is important to note that each form of Al<sub>2</sub>O<sub>3</sub> is characterized by its own value of band gap<sup>22, 23</sup>. From the previously reported studies, the experimental value of band gap for  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is 7.0-8.5 eV<sup>24</sup>, for amorphous alumina i.e. am-Al<sub>2</sub>O<sub>3</sub><sup>25, 26</sup>, the band gap value is 5.0-7.0 eV and for  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>, the value of band gap<sup>27</sup> is higher, i.e. 8.8 eV, which makes it a perfect insulator. The band gap of Al<sub>2</sub>O<sub>3</sub> depends upon the mode of its fabrication. Due

to the high band gap of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> it becomes suitable for microelectronic devices, as a laser host material (e.g., Ruby laser), as a high strength window material, and as a dosimeter<sup>28</sup> using thermoluminescence readout<sup>29</sup>. During the last decade,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> or sapphire has been suggested as an efficient and reproducible dosimeter for both ultra-violet (UV) light and ionization radiation. It is thus important to work further on the structure<sup>30</sup> of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>.

$\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001) surface has p3 symmetry with Al atoms positioned at the centres of the 3-fold axis and oxygen positioned about the 3-fold axis in a hexagonal structure<sup>31,32</sup>. Unlike this,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> has entirely different symmetry and has been described as a spinal similar to a traditional spinal AB<sub>2</sub>O<sub>4</sub> structure. In  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, the oxygen atoms are arranged in a cubic close-packed lattice, while the A and B cations occupy the tetrahedral and octahedral interstitial sites of the lattice<sup>33</sup>. In order to fabricate pure  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> with a high band gap, care must be taken of the surface and structural symmetry of the  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> sample. Raman scattering is one of the most used non-destructive techniques to understand the microscopic properties and also is important to understand the behaviour of lattice vibrations in the molecule<sup>34-37</sup>. It is highly sensitive because of its ultra-sensitive probe which can quantify the smaller defect present in the sample<sup>38-40</sup>. Besides, advanced

technique such as the Raman spectro-microscopy adds to new capabilities of Raman spectroscopy<sup>41,42</sup> as a tool for understanding various aspects in different materials such as bulk materials, nanomaterials, thin films etc. This technique identifies various Raman active vibration modes<sup>43</sup>, namely  $B_{1g}$ ,  $E_g$ , and  $A_{1g}$  for various materials, which confirm the purity and phase of those materials. The other characterization<sup>44</sup> techniques such as X-ray diffraction (XRD), FTIR and PL spectroscopy are also used to determine the phase and surface structure and luminescent properties of the material.

The aim of the present work is to fabricate pure  $\alpha$ - $Al_2O_3$  for various applications. Preparation has been carried out using wet milling slurry method, followed by characterization to confirm the formation of the sample. Room temperature XRD pattern and Raman spectra of the sample confirms the purity, FTIR analysis confirms the presence of required vibrational patterns and SEM reflects the surface morphology.

### Experimental Section

Single phase  $\alpha$ - $Al_2O_3$  has been fabricated by using slurry method, a two-step process. In the first step, slurry solution has been made in water medium using  $\alpha$ -alumina (A-16SG) (ALCOA, USA) with average particle size of 300 nm. Darvan 821A of 40% aqueous solution of ammonium polyacrylate was used as the dispersant. The slurry was prepared in a polypropylene bottle containing two sizes of alumina balls of approximately 2 mm and 10 mm, respectively. The bigger balls fit the length of the bottle and smaller balls fit half of the slurry. Water, dispersant and N-(hydroxymethyl)acrylamide were taken within the polypropylene bottle to prepare 55% alumina powder slurry. To mix and wet the powder uniformly in the bottle, the powder was added in a step by step manner. Mixing of the solution was carried out in the planetary centrifugal mixer for several minutes for de-airing. Finally, ammonium persulfate (initiator) and tetramethylethylenediamine (catalyst) were added to the slurry with vigorous stirring and casted in the petroleum jelly coated Teflon mould at room temperature.

The second step involves drying and sintering process. The mould was covered with glass plate and kept in a preheated oven at 50°C for 2 h. Then cooling was done at 90% humid condition for 24 h to avoid cracking. The material was finally dried in an air-oven at 50°C for 24 h. Binder burnout was done by slow heating to completely remove the organics and

after that sintering was done to produce the ceramic component.

### Results and Discussion

The surface morphology and microstructure of the dried sample were studied using SEM. The SEM images (Fig. 1) show that a uniform crystalline structure has been formed. The top view of the SEM micrograph indicates that the  $\alpha$ - $Al_2O_3$  sample contained a pure crystalline structure containing grey lines. A complete uniform structure can be seen from the Fig. 1a which has been recorded at low magnification  $\sim 10 \mu m$ . To understand and check the microstructure, SEM has also been recorded at high magnification  $\sim 1 \mu m$  (Fig. 1b) in which grey lines of the unit cells can be seen.

Further, phase purity and structural studies on the prepared  $\alpha$ - $Al_2O_3$  sample have been carried out via XRD pattern, FTIR and Raman spectroscopy. The XRD pattern has been recorded at room temperature (300 K) from the diffraction angle ( $2\theta$  values)  $20^\circ$  to  $90^\circ$  which can be seen in Fig. 2a.

The XRD pattern shows diffraction peaks at various  $2\theta$  values corresponding (h k l) planes, which can be identified as originating from the crystalline phase  $\alpha$ - $Al_2O_3$ . The corresponding miller indices (h k l) confirm the phase and plane present in the prepared sample.  $Al_2O_3$  is known to have defects because of the

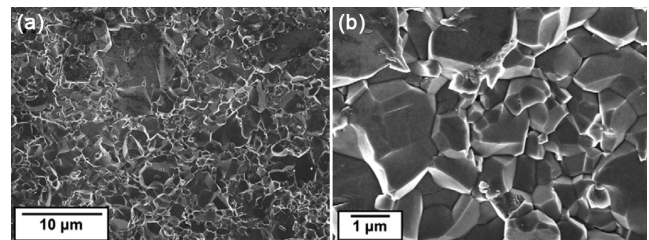


Fig. 1 — (a, b) SEM images of top surface morphology of  $\alpha$ - $Al_2O_3$  at different magnifications

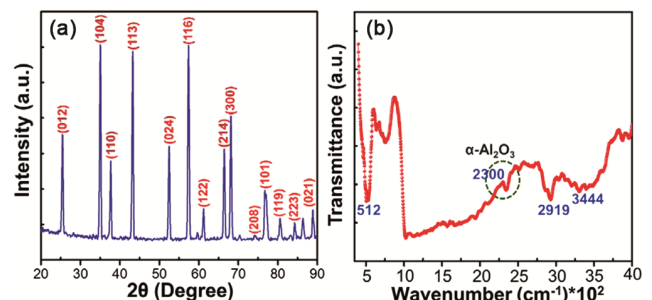


Fig. 2 — (a) X-Ray diffraction pattern and (b) FTIR spectrum of  $\alpha$ - $Al_2O_3$  sample at room temperature

oxygen vacancy, where an  $O^{2-}$  ion moves from a normally occupied lattice site to a vacant interstitial site, which is supposed to be prevalent defect in many oxides. Under thermal conditions, hydrogen atom interacts with the lattice oxygen of  $Al_2O_3$  on the surface, which results in the formation of the oxygen vacancies and changes the surface properties of  $\alpha-Al_2O_3$ . No such defects could be identified through the XRD pattern. Furthermore, no additional peak or hump is seen in the XRD, which reveals that pure single phased  $\alpha-Al_2O_3$  have been fabricated.

The full width at half maximum (FWHM) values at each peak have been determined using ORIGIN software and corresponding crystallite sizes as well as its average value have been calculated using Debye-Scherrer formula (Eq. 1). The data matches the JCPDS-ICDD file number 46-1212. Average crystallite size was found to be 19.91 nm.

$$D = k/\beta\cos \quad \dots(1)$$

where,  $D$  = crystallite size,  $K$  = Scherrer constant (0.89),  $\lambda$  = wavelength (0.154 nm),  $\beta$  = FWHM.

FTIR spectrum (Fig. 2b) shows O-H stretching of  $\alpha-Al_2O_3$  sample at  $3444\text{ cm}^{-1}$ . This stretching occurs only after  $1230^\circ\text{C}$  leading to the formation of the  $\alpha-Al_2O_3$  by the solid state thermally driven transformations from hydroxyl group of aluminium oxide. However, if the annealing temperature is below  $1230^\circ\text{C}$ , the possibility of dehydration level of this hydroxyl group is less, leading to the formation of  $\gamma-Al_2O_3$ . Hence, the degree of the transition to the  $\alpha-Al_2O_3$  or corundum structure depends on the time and the temperature of annealing treatment. Apart from this, the peak for alkyl halide was observed at  $512\text{ cm}^{-1}$  upon verification with the present database and the spectrum matches with  $\alpha-Al_2O_3$ . From Fig. 2b, it can be seen that a peak appearing at  $2300\text{ cm}^{-1}$  corresponds to the  $\alpha-Al_2O_3$  phase, while for  $\gamma-Al_2O_3$ , a broad peak appears around  $2300\text{ cm}^{-1}$  but with less value of transmittance. The FTIR data confirms the purity of the structure and chemical composition of the prepared  $\alpha-Al_2O_3$  sample.

The band gap energy of the sample has been calculated (Fig. 3a) from its absorption spectral studies (Fig. 3b). The  $\lambda_{\text{max}}$  value of the sample in aqueous medium is found to be 265 nm and the band gap energy has been calculated to be 5.47 eV.

To understand the molecular structure of the prepared  $\alpha-Al_2O_3$  crystalline sample, the bond structure diagram and the Raman spectrum have been

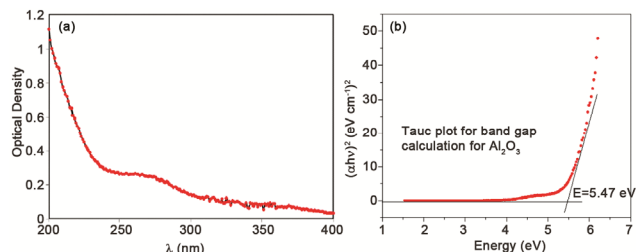


Fig. 3 — (a) Absorption spectrum with  $\lambda_{\text{max}}$  value of 265 nm and (b) Tauc plot for calculation of band gap of prepared alumina sample

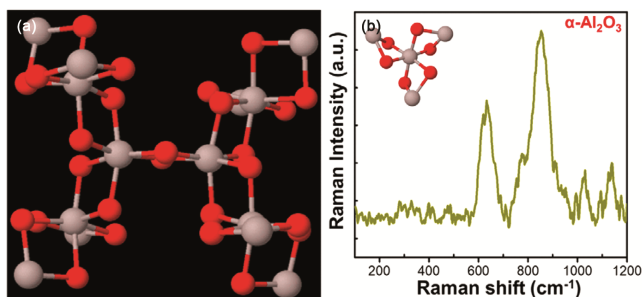


Fig. 4 — (a) Bond structure diagram of alumina using MolView software and (b) Raman spectrum of  $\alpha-Al_2O_3$  at 300K, inset shows the unit bond structure of alumina

presented. The molecular structure of  $\alpha-Al_2O_3$  sample has been generated (Fig. 4a) by using an online software, Mol View. The single phase structure shows the presence of two aluminium atoms attached with three oxygen atoms to complete the vacancies. Due to complexity of the structure, a singular structure of unit cell has been shown in the inset of the Fig. 4b. In the unit cell, grey and red sphere represent the aluminium and oxygen atoms, respectively. To deduce the structure at microscopic level, Raman spectrum of the compound (Fig. 4b) shows two broad Raman active modes at  $650\text{ cm}^{-1}$  and  $875\text{ cm}^{-1}$ . It is well known that in case of  $\gamma-Al_2O_3$ , Raman mode appears at  $630\text{ cm}^{-1}$  and  $700\text{ cm}^{-1}$ . However,  $\gamma-Al_2O_3$  also shows an additional peak compared to  $\alpha-Al_2O_3$ . There are two more Raman active modes present in the  $\alpha-Al_2O_3$  Raman spectrum at  $380\text{ cm}^{-1}$  and  $420\text{ cm}^{-1}$  but these modes are very low intense as compare to the other ones (above mentioned), which can be seen in the spectrum. The structural and spectroscopic analysis confirm that the prepared sample of  $Al_2O_3$  is in single phase and has the  $\alpha$ -form, which is thermodynamically the most stable phase.

## Conclusion

In summary, pure crystalline  $\alpha-Al_2O_3$  has been prepared by wet milling method (slurry method) and

confirmation to this effect is obtained from various well documented characterization techniques. Scanning electron microscopy at different magnifications has been performed to see the top surface morphology of the prepared  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> sample, which confirms the uniformity of the single phase of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> sample. Raman spectroscopy of the material shows two Raman active modes confirming the presence of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>. A fair agreement in all the spectral data has been reached.

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